

# Vibrational Studies of Zinc Antimony Borophosphate Glasses Doped Rare Earth

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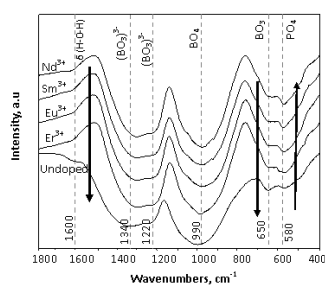
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## Graphical abstract



## Abstract

Zinc antimony borophosphate glasses has been determined for  $[10P_2O_5 - 40B_2O_3 - xSb_2O_3 - (50-x)ZnO]$  and composition of  $[10P_2O_5 - 40B_2O_3 - 40Sb_2O_3 - 10ZnO]$  has been doped with 1 mol% of rare earth (Eu, Nd, Sm, Er). The functions of compositional changes on their structural features were examined using X-Ray Diffraction (XRD) to detect the amorphous phase present. While Fourier Transform Infrared (IR) Spectroscopy were used to identify the presences of vibrational modes and band assignments of phosphate, borate, antimony, zinc and Rare earth ion in the system. XRD results expose that crystalline phase changes with different amount of zinc and antimony substitution. Hydroxyl group absorption also vary due to this composition changes and clearly shown in IR spectroscopy in the ranges  $1400-4000\text{ cm}^{-1}$ . Result of IR spectroscopy indicated that bands around  $1440\text{ cm}^{-1}$  and  $760\text{ cm}^{-1}$  was ascribed to the  $\nu_{as}(B-O-B)$  and  $\nu_s(P-O-P)$  vibration respectively. The changes of this vibration indicated that P-O-B linkage was formed near  $660\text{ cm}^{-1}$ . The modification of zinc antimony borophosphate glasses with rare earth was studied and showed present of rare earth ion in the glass system does not change the structural features.

**Keywords:** Glass; borophosphate; quaternary system; infrared spectroscopy; raman spectroscopy

## Abstrak

Kaca zink antimoni borofosfat ditentukan bagi  $[10P_2O_5 - 40B_2O_3 - xSb_2O_3 - (50-x)ZnO]$  dan pada komposisi  $[10P_2O_5 - 40B_2O_3 - 40Sb_2O_3 - 10ZnO]$  kaca telah didop dengan 1 mol% nadir bumi (Eu, Nd, Sm, Er). Kesan dari perubahan komposisi terhadap ciri-ciri struktur sampel telah diperiksa menggunakan Difraktometer Sinar X-Ray (XRD) untuk mengesan fasa polihabur. Manakala Inframerah (IR) Spektroskopi digunakan untuk mengenalpasti mod getaran dalam kumpulan fosfat, borat, antimoni, zink dan ion nadir bumi di dalam sistem. Keputusan XRD mendapati bahawa fasa kristal berubah seiring perubahan komposisi zink dan antimoni. Perubahan terhadap komposisi juga menunjukkan perubahan terhadap kumpulan hidroksil dan boleh dilihat dengan jelas dalam spektroskopi IR pada julat  $1400-4000\text{ cm}^{-1}$ . Keputusan IR juga menunjukkan bahawa terdapat mod getaran bagi kumpulan vas (BOB) pada frekuensi  $1440\text{ cm}^{-1}$  dan kumpulan  $\nu_s$  (POP) di sekitar  $760\text{ cm}^{-1}$ . Perubahan keamatan mod getaran ini menunjukkan bahawa hubungan P-O-B telah terbentuk berhampiran  $660\text{ cm}^{-1}$ . Pengubahsuaian gelas borofosfat zink antimoni dengan nadir bumi telah dikaji dan menunjukkan kehadiran ion nadir bumi dalam sistem kaca tidak mengubah ciri-ciri struktur.

**Kata kunci:** Kaca; borofosfat; sistem kuantenari; spektroskopi inframerah; spektroskopi raman

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## 1.0 INTRODUCTION

Considering the advantages of borate and phosphate glass, it can be expected to facilitate tailoring of the physical and chemical properties of interest for specific technological application. Phosphate glasses are important material due to their high thermal expansion coefficient and glass transition temperature make it suitable for solid state batteries and technological application of solid state ionic as electrolytes [1]. Besides, borate glasses can be a great hosts for a glass system because they can accommodate large concentrations of active ions and offer an important range of compositional possibilities such as metaborate, pentaborate,

orthoborate, diborate, and pyroborate same as phosphate glasses which also offer an important opportunity to investigate fundamental structural about glass, because their large range of glass forming compositions.

Unfortunately, phosphate and borate alone is not a stable compound for a glass but the chemical durability of phosphate glasses can be improve and stabilize by the incorporation of boron oxide into the structural network as a glass formers and modifier so that borophosphate glasses have acceptable chemical durability compare to pure borate and pure phosphate [2]. The combination of this two glass formers  $P_2O_5$  and  $B_2O_3$  posses a variety of other useful properties. In fact, the presence of  $P_2O_5$  in the borate

glasses improves the glass quality when modified with rare earth [3].

Borate and phosphate glasses doped with various rare earths have been studied mainly due to their non linear optical properties in recent years. Its means only a few researches were carried out about borophosphate glasses dope rare earth. One of the primarily spectral studies of doped rare earth oxide in sodium borophosphate glass has been conducted and the result showed interesting features about the split band. In fact, they only focus on ternary borophosphate system. Therefore, the presence study is done in order to know the structural features of antimony zinc borophosphate ( $\text{ZnO-Sb}_2\text{O}_3\text{-B}_2\text{O}_3\text{-P}_2\text{O}_5$ ) modified by with various rare earth such as  $\text{Er}^{3+}$ ,  $\text{Eu}^{3+}$ ,  $\text{Nd}^{3+}$  and  $\text{Sm}^{3+}$  ions. The effect of replacement of ZnO by  $\text{Sb}_2\text{O}_3$  on the glass structure was also investigated. The information about structural units present in this glass has been obtained from Infrared Spectroscopy and XRD techniques.

## 2.0 EXPERIMENTAL

The glass sample in the series of  $10\text{P}_2\text{O}_5\text{-}40\text{B}_2\text{O}_3\text{-(x)Sb}_2\text{O}_3\text{-(50-x)ZnO}$ : ( $x=0, 10, 20, 30, 40, 50$ ) using  $\text{H}_3\text{PO}_4$ ,  $\text{H}_3\text{BO}_3$  acid,  $\text{Sb}_2\text{O}_3$  and ZnO powder as the raw materials has been successfully prepared by melt quenching technique. Six glass samples were prepared based on 30g batch mixture. However, only one sample with successive physical properties such as high stability and crystal clear borophosphate glass are chosen will be doped with 1% of rare earths such as Eu, Er, Nd, and Sm to study their structure properties.

Based on previous research, they were found that homogeneous glasses were formed on  $\text{P}_2\text{O}_5$  rich side of the system with 0-30 mol%  $\text{B}_2\text{O}_3$  but become white after a few hours due to hygroscopic nature of phosphate [4]. Thus, the content of  $\text{P}_2\text{O}_5$  in the borophosphate glass system for present study was fixed to 10 mol% and  $\text{B}_2\text{O}_3$  was fixed at 40 mol%.

X-ray diffraction (XRD) is used to identify the phase of the glass sample. For this experiment, the amorphous/crystalline phase of the sample was determined using Siemens Diffractometer D5000 system operating at 40kV, 30mA at room temperature in Faculty of Mechanical Universiti Teknologi Malaysia with  $\text{CuK}_\alpha$  radiation ( $\lambda = 1.54 \text{ \AA}$ ). Diffraction pattern were collected at angle  $2\theta$  range from  $10^\circ - 80^\circ$  and the steps size is  $0.050^\circ$  with 2s time per step. In order to reveal the phase compositions, the bulk sample were prepared by grind it into a relatively fine powder form.

Potassium bromide KBr pellet technique was used to study the molecular structure of a glass sample by obtaining their infrared absorption spectra. Bulk sample of glass sample were grind in a clean mortar to a fine powder and has been mix homogenously at ratio of 1:100 with KBr powder before the mixture immediately pressed with a pressure of 5 ton per square inch to yield a approximately thickness of 1.0mm transparent pellet. The samples then examined using a Perkin-Elmer 1710 Fourier Transform Infrared (FTIR) Spectrometer at the Chemical department, Faculty of Science, Universiti Teknologi Malaysia. The examination has done at room temperature and all the infrared spectra of the glasses were record over the range of wavenumber  $1800\text{-}400\text{cm}^{-1}$  using 100 scans at  $4 \text{ cm}^{-1}$  resolution.

## 3.0 RESULTS AND DISCUSSION

Figure 1 represent the XRD pattern of powdered glass samples measured at room temperature. A broaden diffraction lines reveal typical characteristic pattern for amorphous component. Despite

of amorphous component, diffraction line observed for sample containing less than 30 mol%  $\text{Sb}_2\text{O}_3$  prove the presence of additional polycrystalline related to ZnO phase. Comparing X-ray diffraction pattern for this three composition ( $x= 0, 10, 20$ ), it shows that the diffraction lines were at the same position but different intensities.

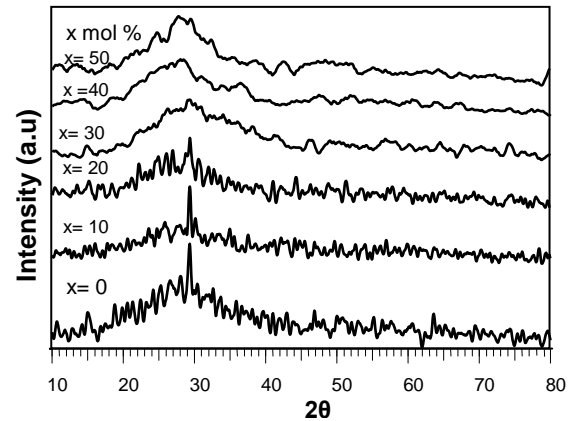


Figure 1 The X-Ray Diffraction (XRD) pattern of  $10\text{P}_2\text{O}_5\text{-}40\text{B}_2\text{O}_3\text{-}x\text{Sb}_2\text{O}_3\text{-(50-x)ZnO}$  glasses

The infrared spectra of zinc antimony borophosphate glasses have focus on their low frequency part for identifying further analyzed the functional group of sample. The IR spectra of  $10\text{P}_2\text{O}_5\text{-}40\text{B}_2\text{O}_3\text{-(x)Sb}_2\text{O}_3\text{-(50-x)ZnO}$  [ $0 \leq x \leq 50 \text{ mol\%}$ ] in the range  $400$  to  $1800 \text{ cm}^{-1}$  are shown in Figure 2.

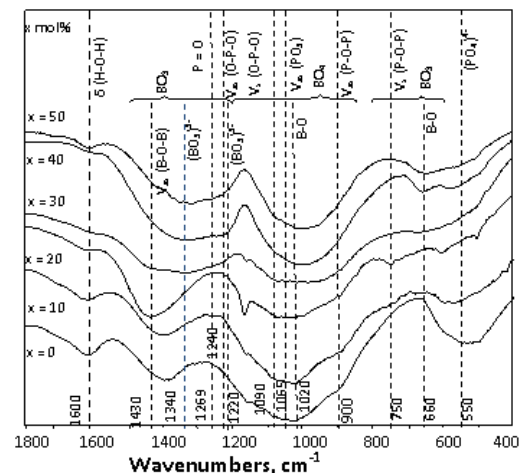


Figure 2 FT-Infrared spectra of  $10\text{P}_2\text{O}_5\text{-}40\text{B}_2\text{O}_3\text{-}x\text{Sb}_2\text{O}_3\text{-(50-x)ZnO}$  glasses in the frequency range of  $400\text{-}1800 \text{ cm}^{-1}$

As modifier, ZnO and  $\text{Sb}_2\text{O}_3$  will changed the borophosphate glasses system absorption peaks when their concentration was increased and decreased respectively. The infrared absorption bands in low frequency associated with the characteristic vibrational modes of borophosphate. The IR bands assigned in this study are based on the work done by other researchers [4][5][6][7][8]. The vibrational modes of pure borate glasses are mainly active in three infrared spectral regions [9]. The absorption peak around  $1200\text{-}1500 \text{ cm}^{-1}$  is assigned to B-O stretching trigonal  $\text{BO}_3$  units, at  $850\text{-}1200 \text{ cm}^{-1}$  is ascribed to the B-O stretching of tetrahedral  $(\text{BO}_4)^-$  units and the binding vibration of various borate segment is identified at  $600\text{-}800 \text{ cm}^{-1}$ .

Four distinct vibrational bands can be characterized for the pure phosphate composition. The vibration at  $1252\text{ cm}^{-1}$  was ascribed to stretching vibration of  $\nu_s(\text{P}=\text{O})$  bond in  $\text{PO}_4$  structural units with one nonbridging oxygen atom, band at  $1206\text{ cm}^{-1}$  is assign to the symmetrical stretching vibrations  $\nu_s(\text{O-P-O})$  group of non bridging oxygen in phosphate chain. The other band at  $793\text{ cm}^{-1}$  is due to symmetrical stretching vibration  $\nu_s(\text{P-O-P})$  of the bridging oxygen atoms connecting linking neighboring  $\text{PO}_4$  tetrahedra and a distinct band at  $500\text{ cm}^{-1}$  is assigned to the deformation vibration of  $\text{PO}_4$  units [10].

Significant changes in the the infrared spectra of  $10\text{P}_2\text{O}_5 - 40\text{B}_2\text{O}_3 - x\text{Sb}_2\text{O}_3 - (50-x)\text{ZnO}$  with the composition ( $0 \leq x \leq 50$ ) in the range of  $400\text{--}1800\text{ cm}^{-1}$  (Figure 2) with increasing content of  $\text{Sb}_2\text{O}_3$  replacing ZnO manifest the incorporation of  $\text{Sb}_2\text{O}_3$  into the structural network of parent borophosphate glass. The spectrum of the parent glass  $10\text{P}_2\text{O}_5 - 40\text{B}_2\text{O}_3 - 50\text{ZnO}$  borophosphate glass shows a broad complex band appears within range  $850\text{--}1200\text{ cm}^{-1}$  is composed of several individual bands ascribed to the B-O stretching vibrations of  $\text{BO}_4$  tetrahedral at  $1020\text{ cm}^{-1}$  and three bands characteristic of P-O stretching vibration of  $\nu_s(\text{O-P-O})$ ,  $\nu_{as}(\text{PO}_3)$  and  $\nu_{as}(\text{P-O-P})$  at  $1090$ ,  $1065$  and  $900\text{ cm}^{-1}$  respectively with all the symmetric and asymmetric stretching vibrations observed in the spectra are characteristic of  $\text{Q}^2$  and  $\text{Q}^1$  groups.

The band at this range obviously become smaller when  $\text{Sb}_2\text{O}_3$  was increased ( $x=50\text{ mol}\%$ ) indicated the change in the coordination of boron atoms from tetrahedral  $\text{BO}_4$  to trigonal  $\text{BO}_3$  units. Several studies justified the incorporation of  $\text{Sb}_2\text{O}_3$  into the borophosphate network creates new non-bridging oxygen atoms, which are at the origin of the different  $\text{BO}_4$  sites [11]. The formation of  $\text{BO}_4$  units into  $\text{BO}_3$  units also can be accompanied by the formation of trigonal bipyramidal  $\text{SbO}_4\text{E}$  (E = 5s2 lone pair of electrons) units. The conversion between  $\text{BO}_3$  units and  $\text{BO}_4$  units can be confirm by the decreases of the band at  $1430$ ,  $1340$  and  $660\text{ cm}^{-1}$  characteristic for stretching vibration of  $\nu_{as}(\text{B-O-B})$ , B-O bond of  $(\text{BO}_3)^{3-}$  units in metaborates, pyroborates also ortoborates, and B-O linkage (B-O-B) bond in the borate  $\text{BO}_3$  network respectively.

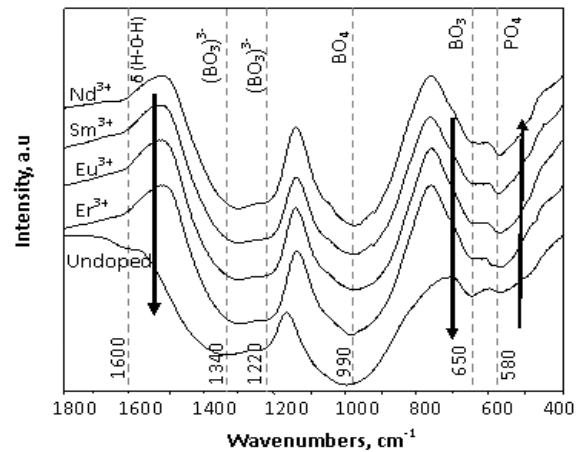
With increasing  $\text{Sb}_2\text{O}_3$  content, the symmetric stretching vibration of non bridging phosphate atoms in  $\text{PO}_4$  units at  $550\text{ cm}^{-1}$  assigned to  $\text{Q}^0$ , orthophosphate group gradually decrease. This result shows, at higher concentration of ZnO, a broad band within range  $900\text{--}1200\text{ cm}^{-1}$  is owing to stretching vibration of  $\nu_{as}(\text{O-P-O})$  at  $1240\text{ cm}^{-1}$  and  $\nu_{as}(\text{PO}_3)$  at  $1065\text{ cm}^{-1}$ . Both of this vibration is characteristic of the stretching vibration of metaphosphate  $\text{Q}^2$  and diphosphate  $\text{Q}^1$  respectively. An incorporation of ZnO units into  $\text{P}_2\text{O}_5 - \text{B}_2\text{O}_3 - \text{Sb}_2\text{O}_3$  glasses results in the depolymerisation of phosphate chains. The process of breaking P-O-P bonds to isolated  $\text{PO}_4$  units is corresponds to the presence of  $\text{Q}^1$  and  $\text{Q}^2$  unit in phosphate network.

We assume that, the number of P-O-B bonds increase when the number of P-O-P and B-O-B bonds decrease with increasing  $\text{Sb}_2\text{O}_3$  content in the borophosphate glasses due to increased number of  $\text{BO}_3$  unit. In this way, the shift of the stretching vibration of  $\nu_{as}(\text{B-O-B})$  band at the pure Zn borophosphate glasses  $50\text{ZnO}-10\text{P}_2\text{O}_5-40\text{B}_2\text{O}_3$  from  $1400\text{ cm}^{-1}$  slightly to higher frequencies can be explain. The band at  $1400\text{ cm}^{-1}$  can be assigned to the asymmetric stretching vibration of bridging oxygen in borate unit in the glass of this composition.

With replacement of ZnO by  $\text{Sb}_2\text{O}_3$  the further amount of the cation with higher cationic field strength shifts this band from  $1400\text{ cm}^{-1}$  slightly to higher frequency but decrease the number of this band and new band was observed at  $1220\text{ cm}^{-1}$  assigned to stretching vibration of B-O bond of  $(\text{BO}_3)^{3-}$  units involving mainly linkage oxygen connecting different group. Band at  $1340\text{ cm}^{-1}$  corresponding to stretching vibrations of B-O bond of  $(\text{BO}_3)^{3-}$

units in metaborates, pyroborates and ortoborates also arise at higher content of  $\text{Sb}_2\text{O}_3$  borophosphate glasses.

The IR patterns on antimony zinc borophosphate for both doped and undoped glass samples was shows in Figure 3. Borophosphate glasses doped with various rare earths such as  $\text{Nd}^{3+}$ ,  $\text{Sm}^{3+}$ ,  $\text{Er}^{3+}$  and  $\text{Eu}^{3+}$  at the same concentration (1 mol %) were prepared and examined by IR Spectroscopy. The relationship between effects of different type of rare earth in the glass system was study due to their structure in the same host matrices.



**Figure 3** FT-Infrared spectra of and  $10\text{P}_2\text{O}_5 - 40\text{B}_2\text{O}_3 - 40\text{Sb}_2\text{O}_3 - 10\text{ZnO} - 1\text{ RE}$  glasses in the frequency range of  $400\text{--}1800\text{ cm}^{-1}$

By comparing all of the spectra (Figure 3), it clearly shows no intense changes in the finger print region. It is observed that whole spectrum of each rare earth doped glasses seems to be repetitive to that obtained from base undoped glass. In the sample containing rare earth ion, it is observed that the presence of rare earth ions have only minor effect on the IR spectra of studied glasses. This is maybe due to the low content of dopant, so the structural groups remain unchanged giving their characteristic vibration. A distinct band at  $650\text{ cm}^{-1}$  corresponding to B-O linkage (B-O-B) bond in the borate  $\text{BO}_3$  network slightly decrease with ionic radii increasing from Er, Eu, Sm to Nd of rare earth series. Previous study by Pisarski on lead borate glasses indicates that rare earth reduce trigonal  $\text{BO}_3$  units and suggest that the smaller heavier lanthanides have the most effect on the local structure compare to larger lighter rare earth ones [12]. Completely different situation was observed for band at  $500\text{ cm}^{-1}$  due to the deformation vibration of  $\text{PO}_4$  units.

Band at  $1600\text{ cm}^{-1}$  is attributed to the deformation modes of O-H ( $\delta\text{ H-O-H}$ ) and usually indicates the water incorporation before glass devitrification unexpectedly decrease with addition of rare earth ions into the glass system even all doped and undoped sample were prepared under the same processing condition such as sample thickness and melting temperature during glass preparation. In particular, sample under investigation does not have  $\text{H}_2\text{O}$  as unit in the glass network but the presence of these bands causing due to reabsorption of water from atmospheric moisture by pellet sample because the infrared spectra were not recorded in situ [6]. In comparison, glass doped  $\text{Er}^{3+}$  sample shows the most enormously changes towards the OH group. The structural changes proved by the infrared spectroscopy measurement suggest that  $\text{Er}^{3+}$  ions also can improve the borophosphate glass network by reduces the OH group concentration and play role as modifier in zinc antimony borophosphate glasses at low concentration.

#### ■4.0 CONCLUSION

The X-Ray Diffraction and Infrared Spectroscopy studies on the network structure of  $P_2O_5 - B_2O_3 - Sb_2O_3 - ZnO$  glass system doped with various rare earth such as  $Er^{3+}$ ,  $Eu^{3+}$ ,  $Nd^{3+}$  and  $Sm^{3+}$  ions revealed the following discussion: The XRD results of the quaternary samples reveal a change of glass structure and polycrystalline phase was observed for sample containing more than 30 mol% ZnO. At higher ZnO glasses, most of basic coupling unit of P-O bond will affected and the infrared spectra shows that  $Sb^{3+}$  enters the glass network by breaking up the B-O-B at higher  $Sb_2O_3$  content. Decreases absorption  $\nu_{as}(B-O-B)$  and  $\nu_s(P-O-P)$  vibration bands indicated that P-O-B linkage was formed near  $660\text{ cm}^{-1}$ . Hygroscopic nature of borophosphate glass were minimum at high antimony content and introduction Erbium ions into the zinc antimony borophosphate glass show the most enormously changes towards moisture sensitivity compare to other rare earth ions.

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